

# Pyrite FeS<sub>2</sub> Kickoff Meeting

August 3, 2010

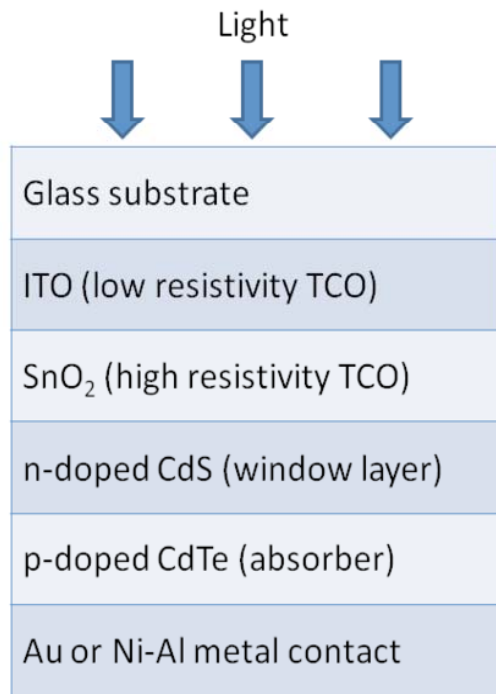


# Meeting schedule

9:00	Intro to pyrite, collaboration	Matt Law
9:20	Phase Field Crystal (PFC) modeling	John Lowengrub
9:40	CVD thin film growth	Nick Berry
9:55	XPS analysis of pyrite films	Ming Cheng
10:10	Colloidal pyrite nanocrystals and films	James Puthussery
10:25	Colloidal & molecular approaches	Sean Seefeld
10:40	New ligands / Molecular approaches	Amanda Weber
10:55-12:00	Discussion and planning	



# First Solar – CdTe thin film company



- ★ Largest thin film manufacturer
- ★ 1.25 GW/yr manufacturing capacity in 2010, projected 1.8 GW/yr in 2012
- ★ \$0.85/watt module manufacturing cost (3Q 2009), versus crystalline silicon which is closer to \$3/watt
- ★ First Solar's average module efficiency is 11% (3Q 2009)
- ★ Best CdTe cell efficiency is 16.5% (NREL)
- ★ Module cost < 50% total installed cost

40-MW installation  
in Waldpolenz,  
Germany



**Scalable to TWs?**

# Will tellurium limit scalability of CdTe PV?

U.S. Geological Survey, Mineral  
Commodity Summaries, January 2008

World Refinery Production, Reserves, and Reserve Base:

	Refinery production		Reserves <sup>3</sup>	Reserve base <sup>3</sup>
	2006	2007 <sup>e</sup>		
United States	W	W	3,000	6,000
Canada	75	75	700	1,500
Japan	24	25	NA	NA
Peru	33	35	1,600	2,800
Other countries <sup>4</sup>	NA	NA	16,000	37,000
World total (rounded)	<sup>5</sup> 132	<sup>5</sup> 135	21,000	47,000

World Resources: The figures shown for reserves and reserve base include only tellurium contained in economic copper deposits. These estimates assume that less than one-half of the tellurium contained in unrefined copper anodes is actually recovered.

- 135 tons Te produced in 2007
  - 1 GW CdTe PV uses ~90 tons Te
  - 1 TW CdTe would require entire worldwide reserve base
- ... More Te will have to be found/mined economically



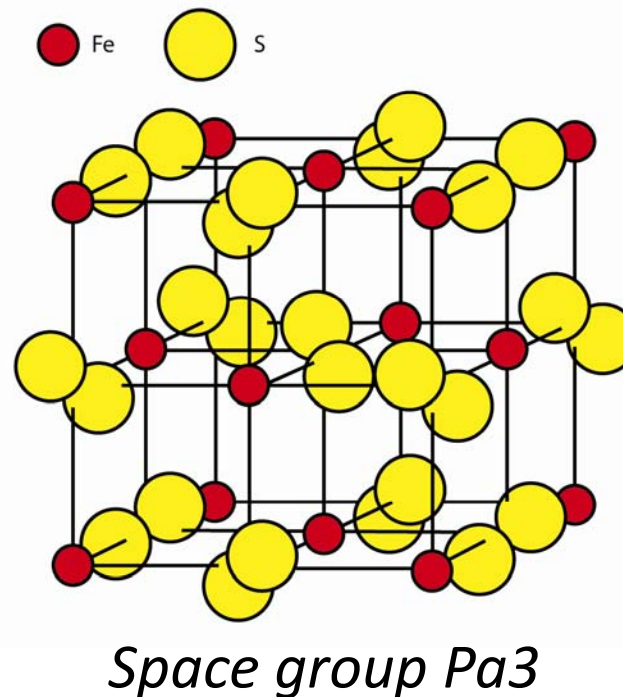
# Iron Pyrite ( $\text{FeS}_2$ )

## Fool's gold

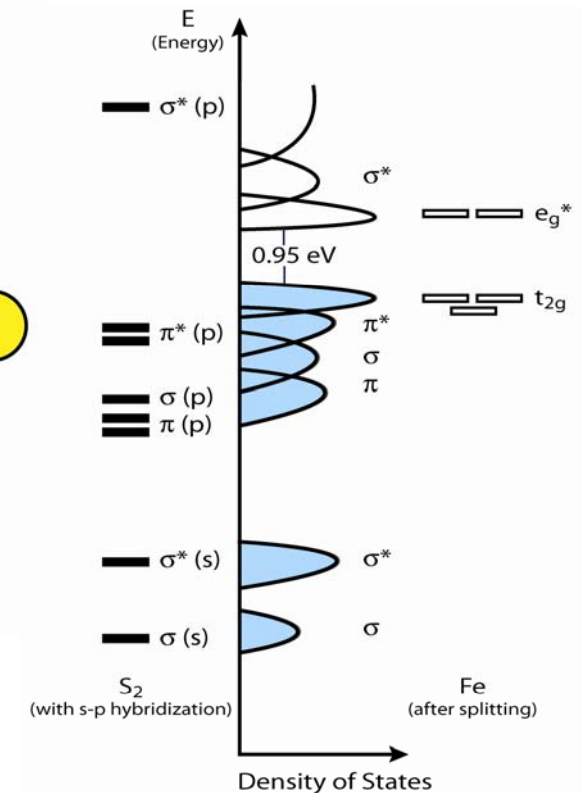


- suitable bandgap (0.95 eV)
- very strong light absorption
- adequate diffusion lengths
- extremely cheap
- infinitely abundant (most abundant sulfide in crust)

## Crystal structure



## Electronic structure



Pyrite tops a short list of thin film materials capable of scaling to multiple TWs without resource limitations

# Basic properties

**Table 1. Pyrite Properties**

Bandgap	$E_g = 0.95$ eV, indirect
Absorption coefficient	$> 10^5$ cm <sup>-1</sup> for $h\nu > 1.3$ eV
Electron effective mass	$0.25m_e$
Hole effective mass	$(2.2 \pm 0.7)m_e$
Electron mobility	$\mu_{\max} = 300$ cm <sup>2</sup> V <sup>-1</sup> s <sup>-1</sup> $\mu_{\min} = 10$ cm <sup>2</sup> V <sup>-1</sup> s <sup>-1</sup>
Hole mobility <sup>68</sup>	$\mu_{\max} = 200$ cm <sup>2</sup> V <sup>-1</sup> s <sup>-1</sup> $\mu_{\min} = 0.02$ cm <sup>2</sup> V <sup>-1</sup> s <sup>-1</sup>
Melting point	1016 K (decomposes to Fe <sub>1-x</sub> S + ½S <sub>2</sub> )
Minority carrier diffusion length	0.12-1.0 μm
Flat-band work function	4.3 eV
Dielectric constant	$\epsilon = 10.9$
Average refractive index	4.5

Data obtained from Reference 8 except as indicated.

# Optical absorptivity

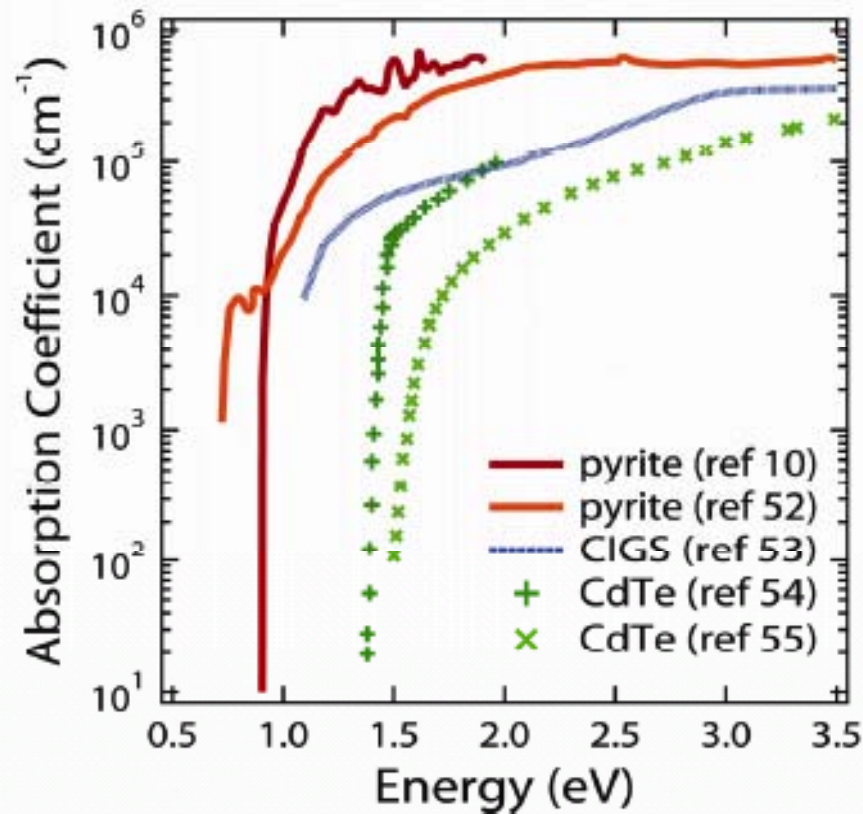


Figure 2.1. Optical absorption coefficients of pyrite, CIGS, and CdTe

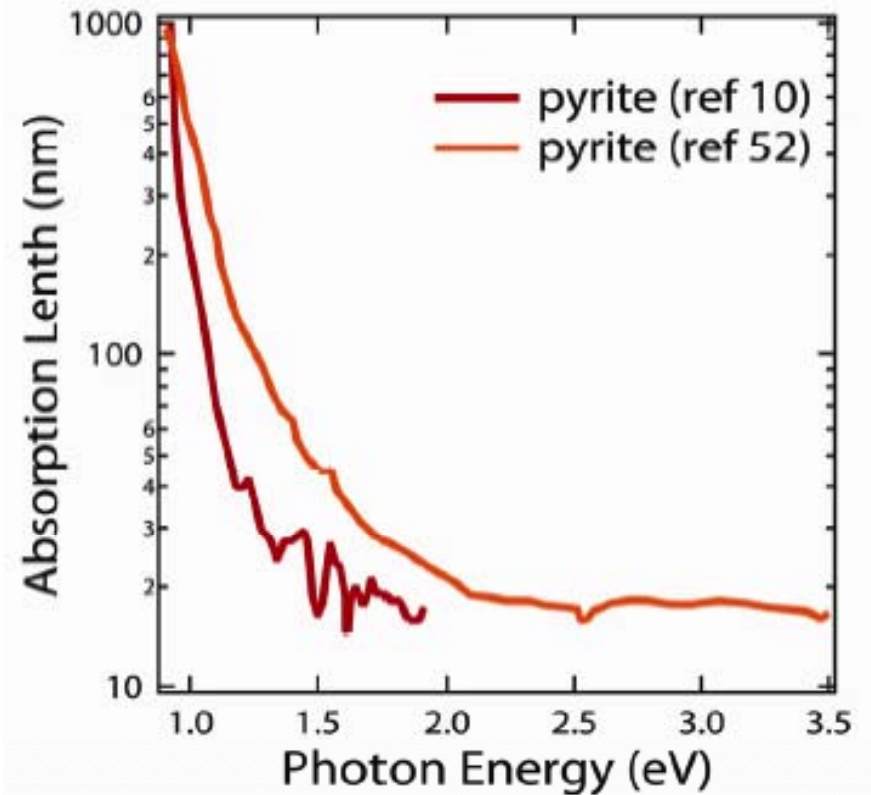


Figure 2.6. Optical absorption length of pyrite films.



# Previous work

Helmut Tributsch (Hahn-Meitner Institut) pioneered pyrite for PV starting in 1983

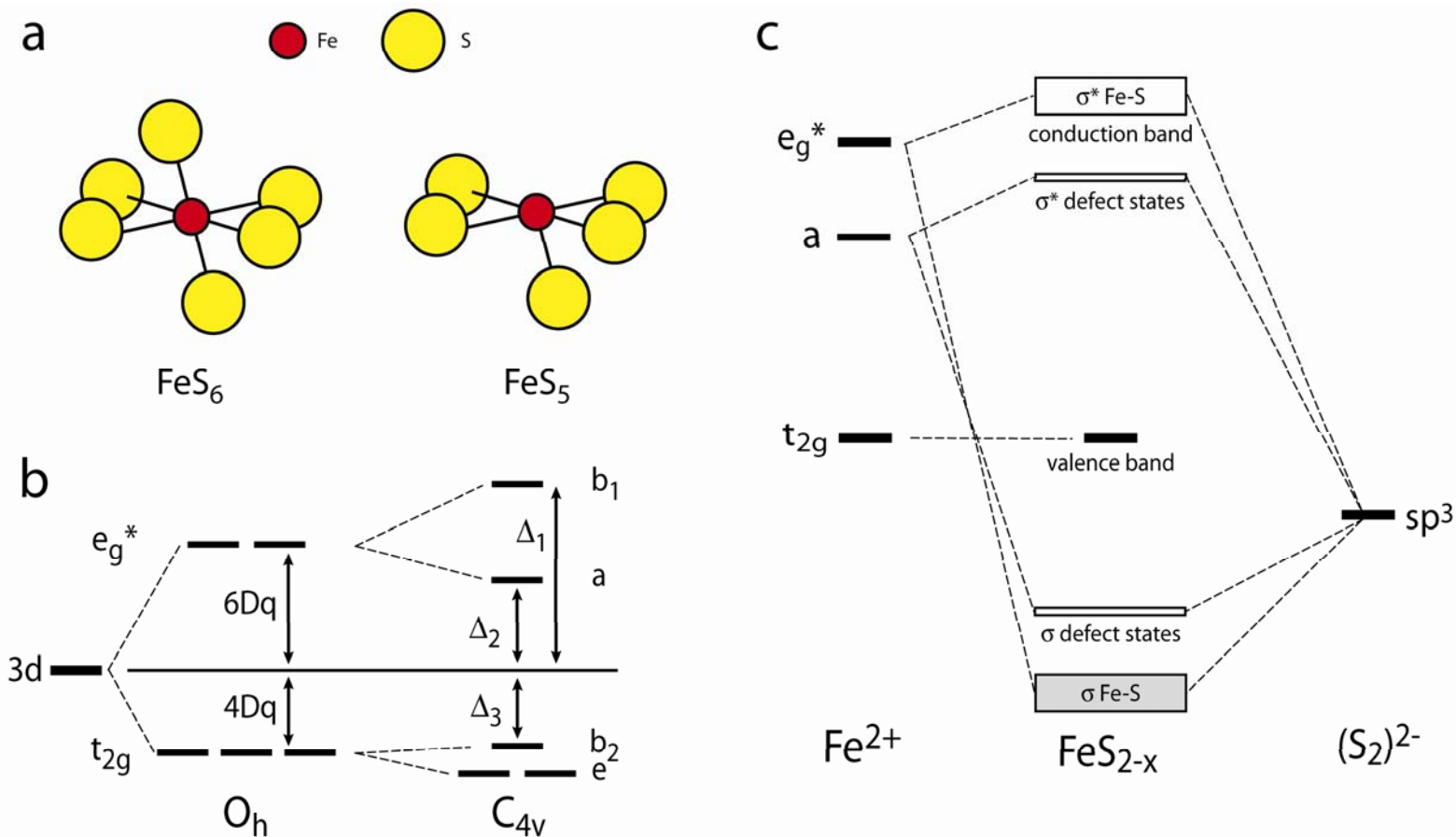
Table 2. Compilation of Pyrite Solar Cell Reports

Year	Cell Type	Device Structure	Performance	Reference
1984	photoelectrochemical	<i>n</i> -FeS <sub>2</sub> single crystal aqueous I <sup>-</sup> /I <sub>3</sub> <sup>-</sup> electrolyte	$V_{OC} = 200 \text{ mV}$ $\eta \approx 1\%$	9,63,60,61,69
1984	solid-state Schottky	<i>n</i> -FeS <sub>2</sub> with Ni or Au	$V_{OC} = 100 \text{ mV}$ $\eta < 1\%$	63
1990	photoelectrochemical	<i>n</i> -FeS <sub>2</sub> single crystal aqueous I <sup>-</sup> /I <sub>3</sub> <sup>-</sup> electrolyte	$J_{SC} = 42 \text{ mA cm}^{-2}$ $V_{OC} \approx 200 \text{ mV}$ $\eta = 2.8\%$	56
1991	photoelectrochemical	polycrystalline film heated in H <sub>2</sub> aqueous I <sup>-</sup> /I <sub>3</sub> <sup>-</sup> electrolyte	$J_{SC} = 18.4 \text{ mA cm}^{-2}$ $V_{OC} = 460 \text{ mV}$ $\eta = 3.3\%$	64
1992	solid-state Schottky	<i>n</i> -FeS <sub>2</sub> with Pt, Au, Nb	$J_{SC} = 30 \text{ mA cm}^{-2}$ $V_{OC} = 100 \text{ mV}$ $\eta < 1\%$	70
1992	sensitized TiO <sub>2</sub> photoelectrochemical	thin MOCVD FeS <sub>2</sub> layer on nanocrystalline TiO <sub>2</sub> aqueous [Fe(CN) <sub>6</sub> ] <sup>3-/4-</sup> or I <sup>-</sup> /I <sub>3</sub> <sup>-</sup> electrolyte	$V_{OC} = 600 \text{ mV}$ (from DSSC design) $\eta < 1\%$	65
1995	<i>p-n</i> homojunction	-	$\eta < 1\%$	66
2009	polymer hybrid	bulk heterojunction	$\eta = 0.16\%$	71

# Pyrite suffers from a low voltage

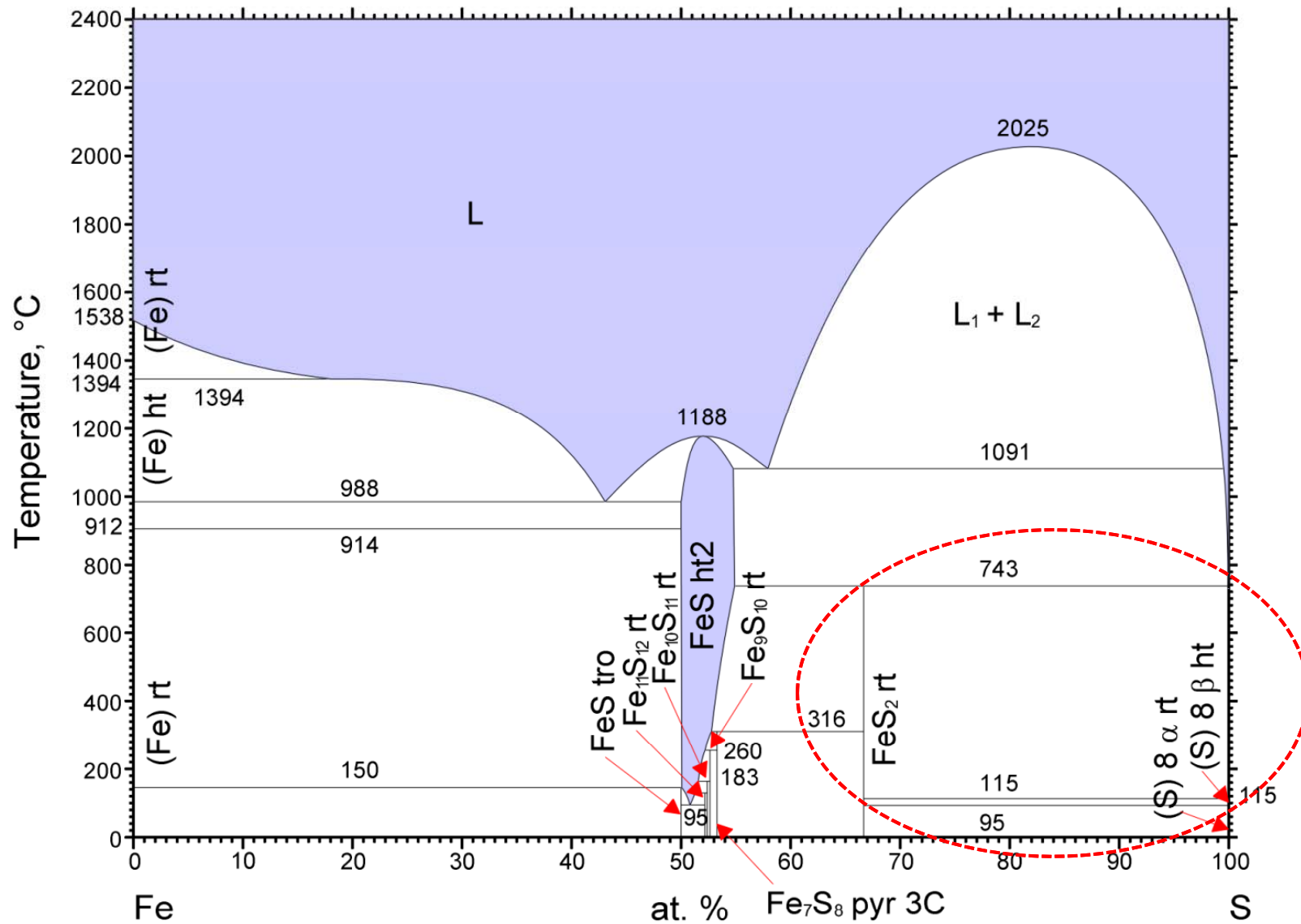
Pyrite tends to be sulfur deficient, and sulfur defects create states in the bandgap, limiting the voltage

Performance:  $40 \text{ mA cm}^{-2}$ ,  $\sim 0.2 \text{ V}$ , 3.3%



Passivating bulk/surface defects is the key to boosting cell efficiency

# Fe-S phase diagram



# Common Fe-S phases

**Table 2.3: Common Crystalline Phases of the Iron-Sulfur System**

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Pyrite (FeS <sub>2</sub> )	$E_g = 0.95$ eV (indirect) Van Vleck paramagnetic semiconductor; cubic
Marcasite (FeS <sub>2</sub> ) <sup>79,80</sup>	$E_g = 0.4$ eV (indirect) diamagnetic semiconductor; orthorhombic
Greigite (Fe <sub>3</sub> S <sub>4</sub> ) <sup>81,82</sup>	$E_g = 0.4$ eV ferrimagnetic semimetal; cubic
Pyrrhotite (Fe <sub>1-x</sub> S, with $x \leq 0.2$ ) <sup>83</sup>	$E_g \approx 0.2$ eV diamagnetic or ferromagnetic semiconductor; hexagonal or monoclinic
Troilite (FeS) <sup>84,85,86</sup>	$E_g = 0.04$ eV antiferromagnetic semiconductor; hexagonal

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# Science Summary

- Pyrite is extremely promising as a scalable PV material. It deserves another look, using modern techniques.
- The main limitation on the efficiency of pyrite cells is their low open-circuit voltage.
- The low voltage is probably caused by iron-derived gap states.
- The most promising way to improve the pyrite photovoltage is to passivate under-coordinated iron ions, especially at the crystal surface.



# Funding: NSF's SOLAR Program

## Synopsis

“The purpose of the CHE-DMR-DMS Solar Energy Initiative is to support interdisciplinary efforts by groups of researchers to address the scientific challenges of highly efficient harvesting, conversion, and storage of solar energy. Groups must include three or more co-Principal Investigators of whom one must be a researcher in chemistry, a second in materials, and a third in mathematical sciences in areas supported by the Divisions of Chemistry, Materials Research, and Mathematical Sciences, respectively. The intent is to encourage new collaborations **in which the mathematical sciences are linked in a synergistic way** with the chemical and materials sciences to develop novel, potentially transformative approaches in an area of much activity but largely incremental advances. Successful proposals will offer potentially transformative projects, new concepts, and interdisciplinary education through research involvement based on the integrated expertise and synergy from the three disciplinary communities.”

**NSF's emphasis is the math effort: use emerging techniques in math to help mat. sci. and chemistry solve solar problems**

# SOLAR 2010

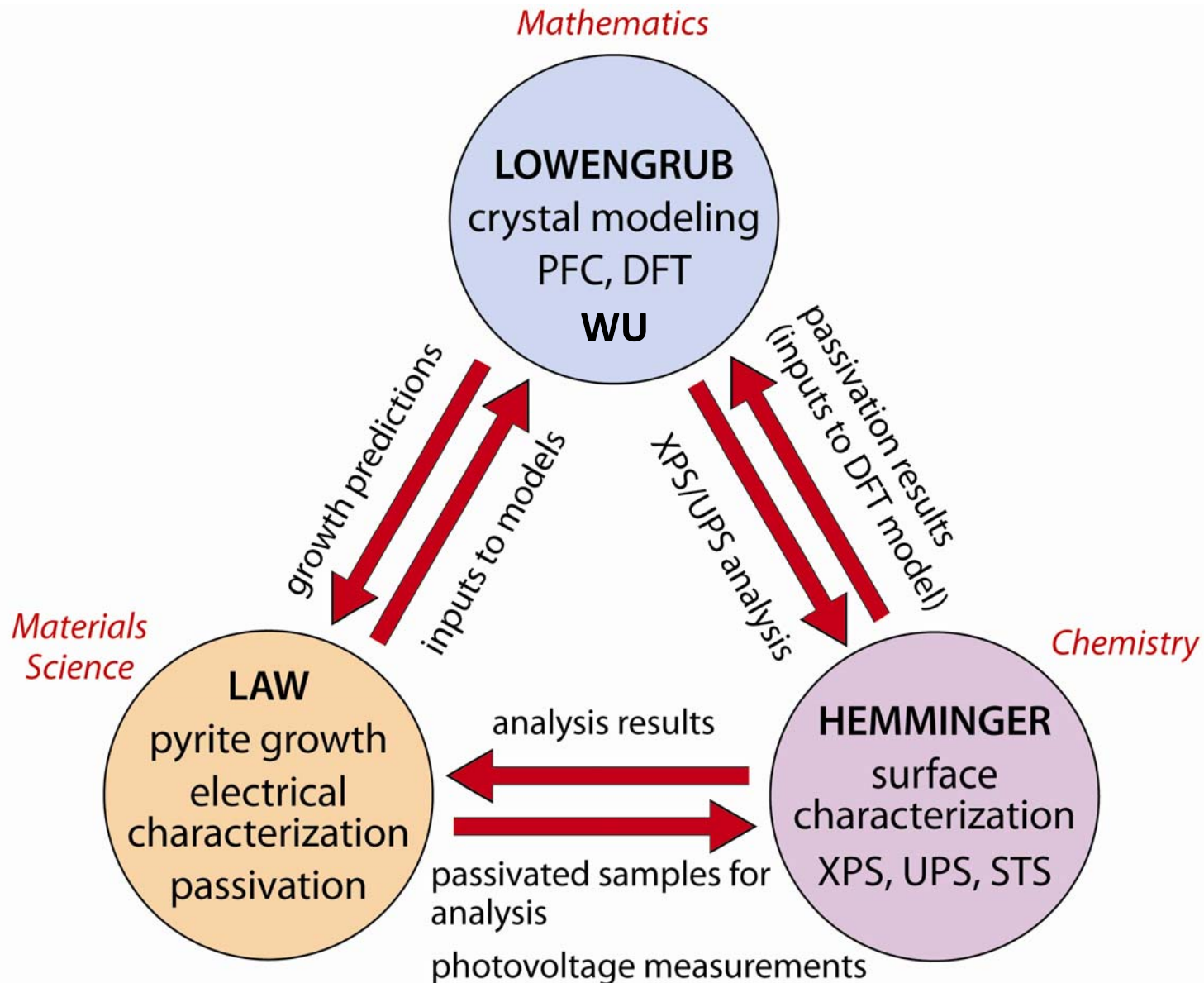
- 3<sup>rd</sup> year of the SOLAR program
- 109 pre-proposals; 30 full proposals; 9 funded
- Our proposal was ranked #1
- We benefitted by not being an organic bulk heterojunction proposal

# Collaboration overview

- Objective: Combine PFC and DFT modeling with pyrite growth and surface characterization to
  - 1) grow device-quality pyrite thin films by 2 methods
  - 2) triple the pyrite photovoltage via passivation
- Law, Hemminger, Lowengrub, and Wu groups.
- 3 years, starting 9/1/10. Project is renewable.
- Funding for 5 students & postdocs.
  - 2 synthesis/characterization (Law)
  - 1 surface characterization (Hemminger)
  - 2 modeling (Lowengrub & Wu)

# Collaboration activities

Model • Grow • Characterize



# Modeling objectives

## Angstrom-to-micron capabilities

### PFC

*Ideally*, show how to grow large grain, stoichiometric pyrite thin films.

More realistically, help to:

- control gas-phase pyrite nucleation
- understand crystal growth
- predict crystal phase
- identify conditions for sintering
- understand SS diffusion
- improve adhesion to substrate
- eliminate voids
- minimize sulfur deficiency
- identify conditions for epitaxy

### DFT

*Ideally*, show how to passivate bulk/surface states to boost voltage.

More realistically, help to:

- identify origin of gap states
- calculate binding energies/geometries
- calculate formation energies
- vet promising surface treatments
- interpret/predict XPS results
- interpret/predict STS results
- simulate FTIR data
- discriminate bulk/surface effects
- understand doping



# Technical challenges

1. Develop and validate a PFC model for FCC pyrite
2. Develop a DFT model of pyrite (bulk & surface)

With input from the models,

3. Produce high-quality pyrite films from pyrite nanocrystal paint.
4. Make high-quality pyrite films by CVD.
5. Grow epitaxial layers on silicon as model systems.
6. Develop surface passivation treatments that increase the pyrite photovoltage  $> 500$  mV.
7. Establish causation between passivation and enhanced electronic performance. Maximize robustness of passivation.
8. By the end of the project, fabricate prototype pyrite p-n heterojunction solar cells with improved voltage and conversion efficiency.

# Surface passivation strategies

- Moderate-temperature annealing in  $S_2$ ,  $H_2S$ , and  $H_2$  atmospheres.
- Coordination of surface iron with strongly-bound organic and inorganic ligands.

Testing with XPS, STS, and photovoltage measurements

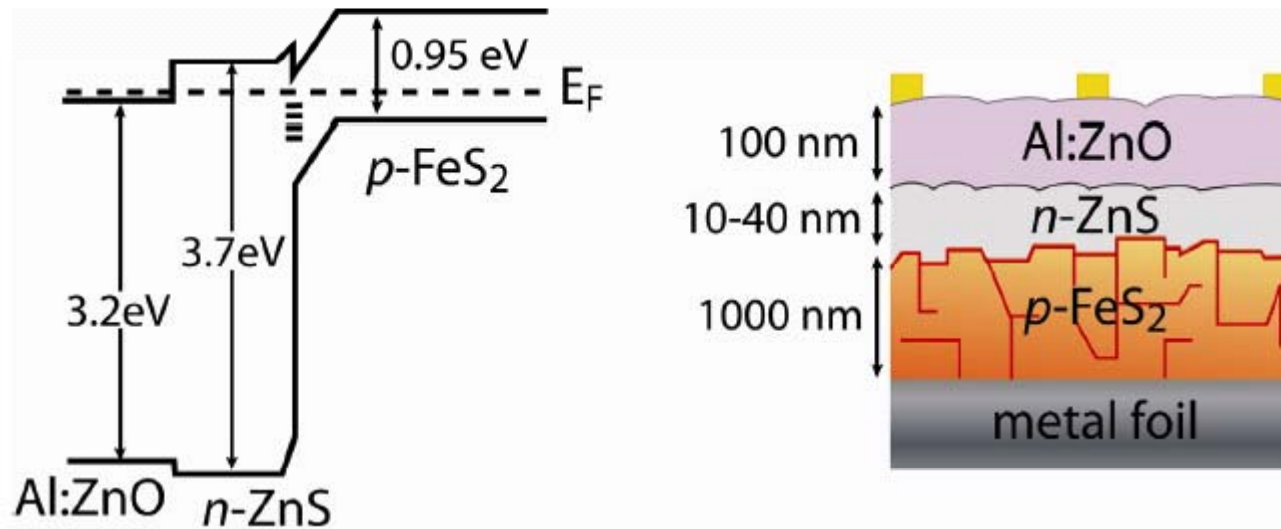
# Project schedule

Year 1: Construct PFC and DFT models and use to  
optimize film microstructure  
tune passivation treatments  
interpret XPS/UPS/STS data

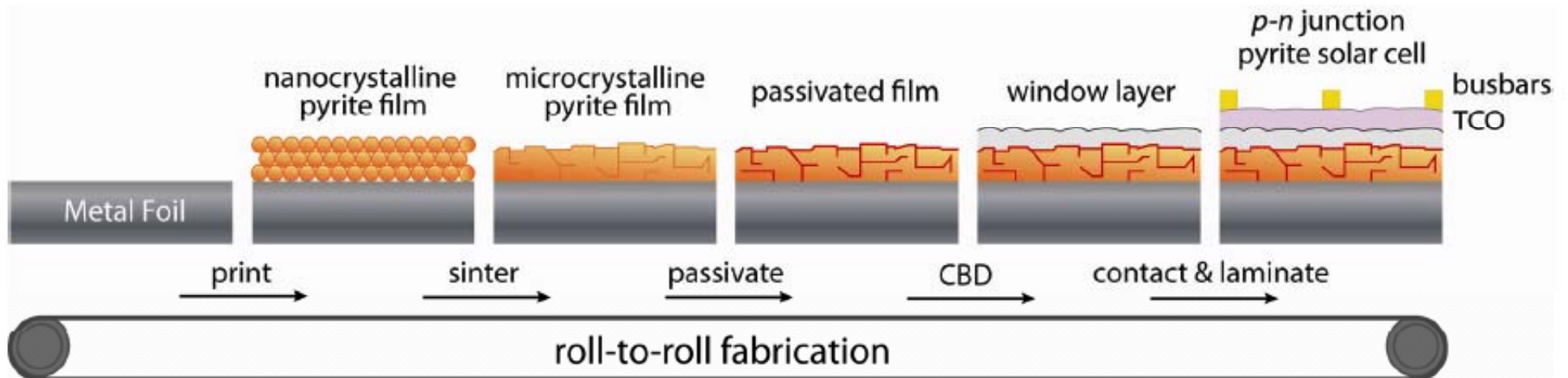
Year 2: Continuation

Year 3: Be producing high-quality pyrite via both synthetic approaches. Establish the connection between microstructure, gap states, and photovoltage. By the end, make record-performance devices.

# End game



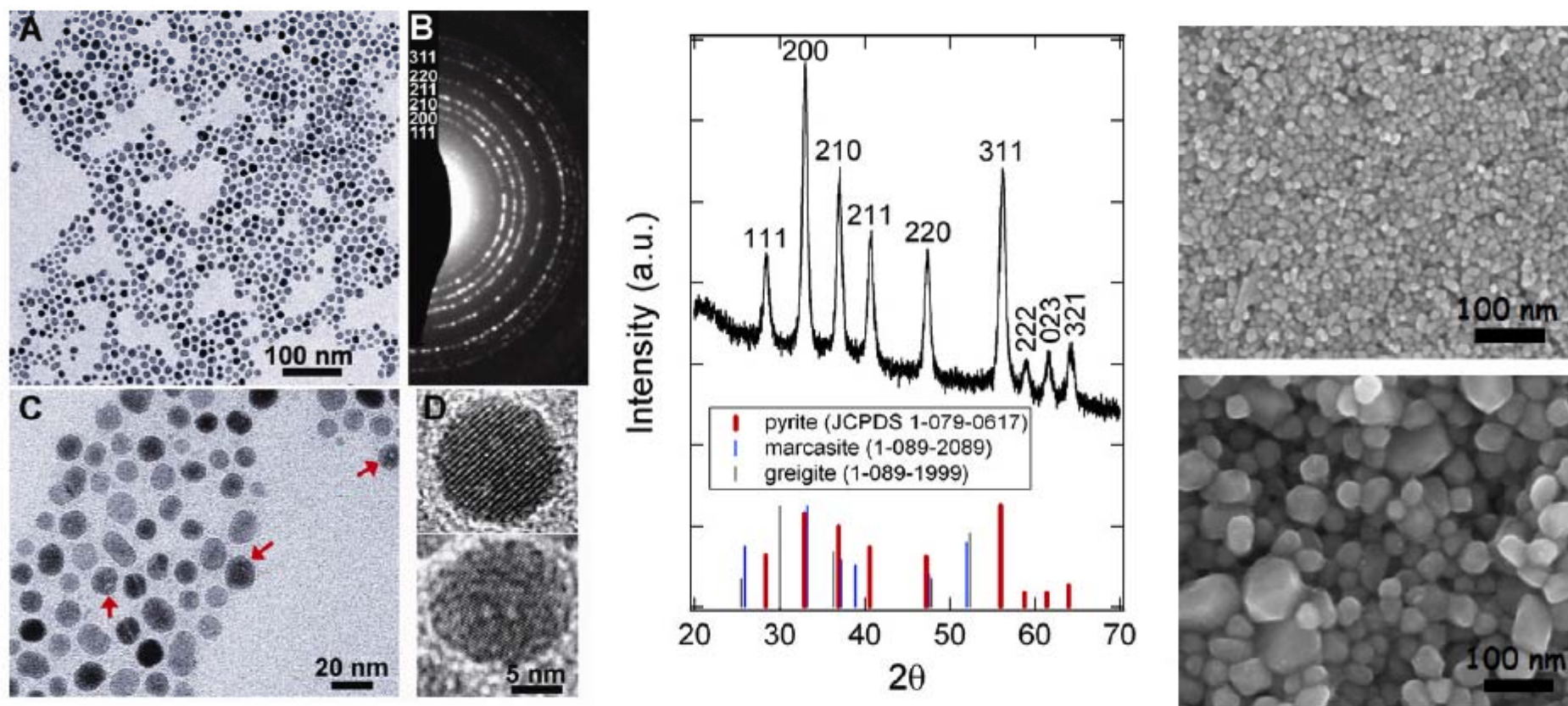
**Figure 2.7.** (left) Band diagram of the heterojunction cell. (right) The device structure.



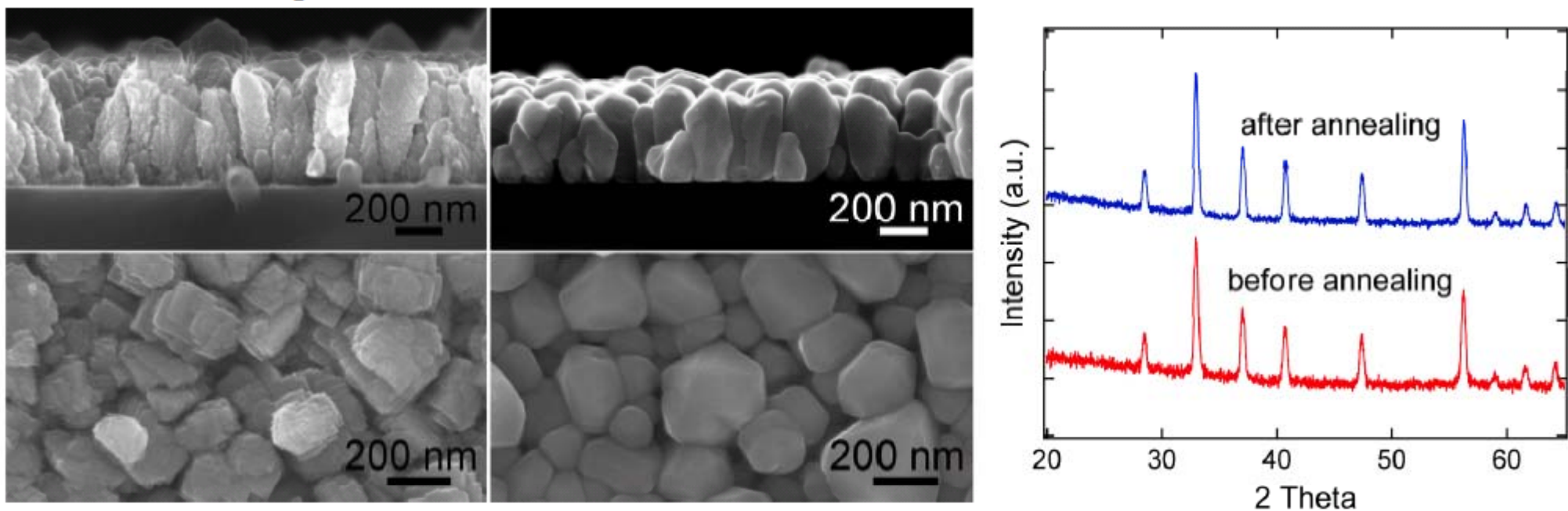
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**Figure 6. Preliminary data.** (*left, middle*) Characterization of colloidal pyrite nanocrystals. (*right*) A pyrite nanocrystal film before (*top*) and after (*bottom*) sintering at 500°C in S<sub>2</sub> vapor.

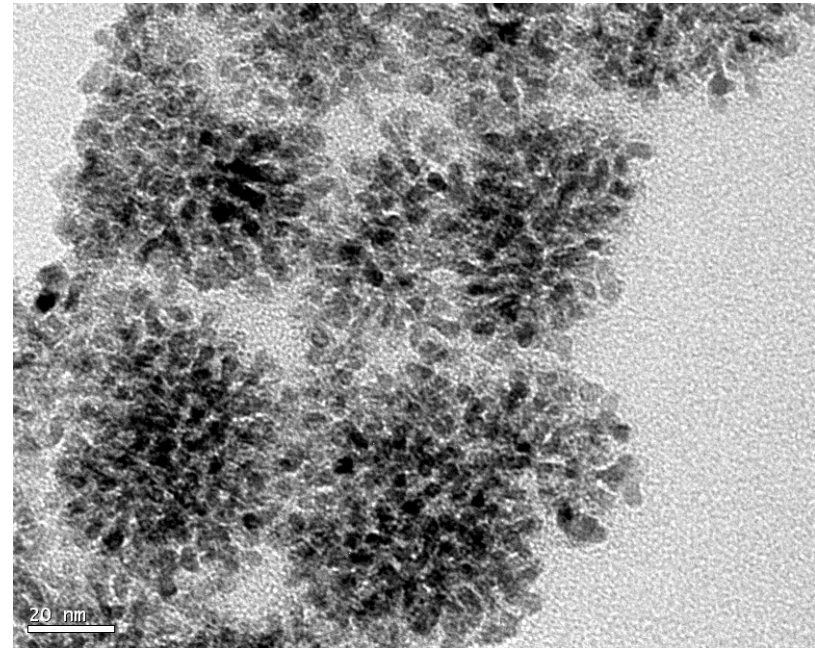


**Figure 7. Preliminary data.** SEM images of a pyrite thin film grown by CVD on glass before (*left*) and after (*middle*) sintering at 500°C in S<sub>2</sub> vapor. XRDs (*right*) show the film remains pure pyrite.

# Synthesizing pyrite NCs is a challenge

- multiple Fe-S phases
- difficult to crystallize
- tends to aggregate

Early results: aggregates



## Iron sulfide phases

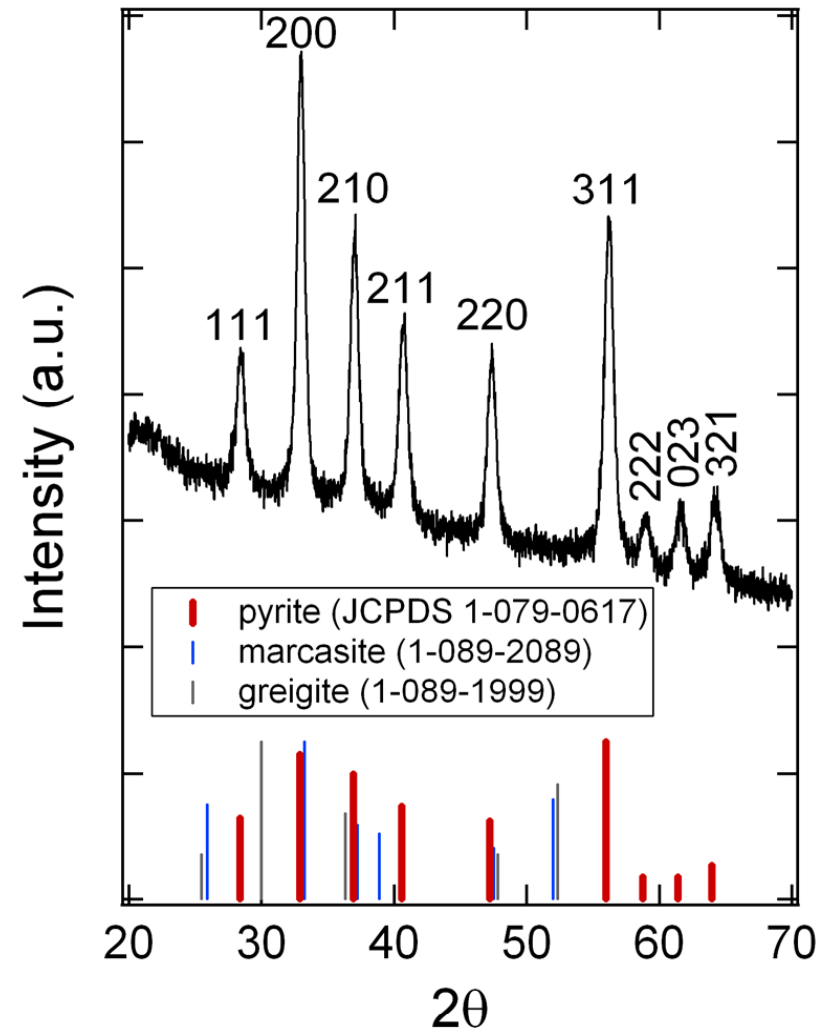
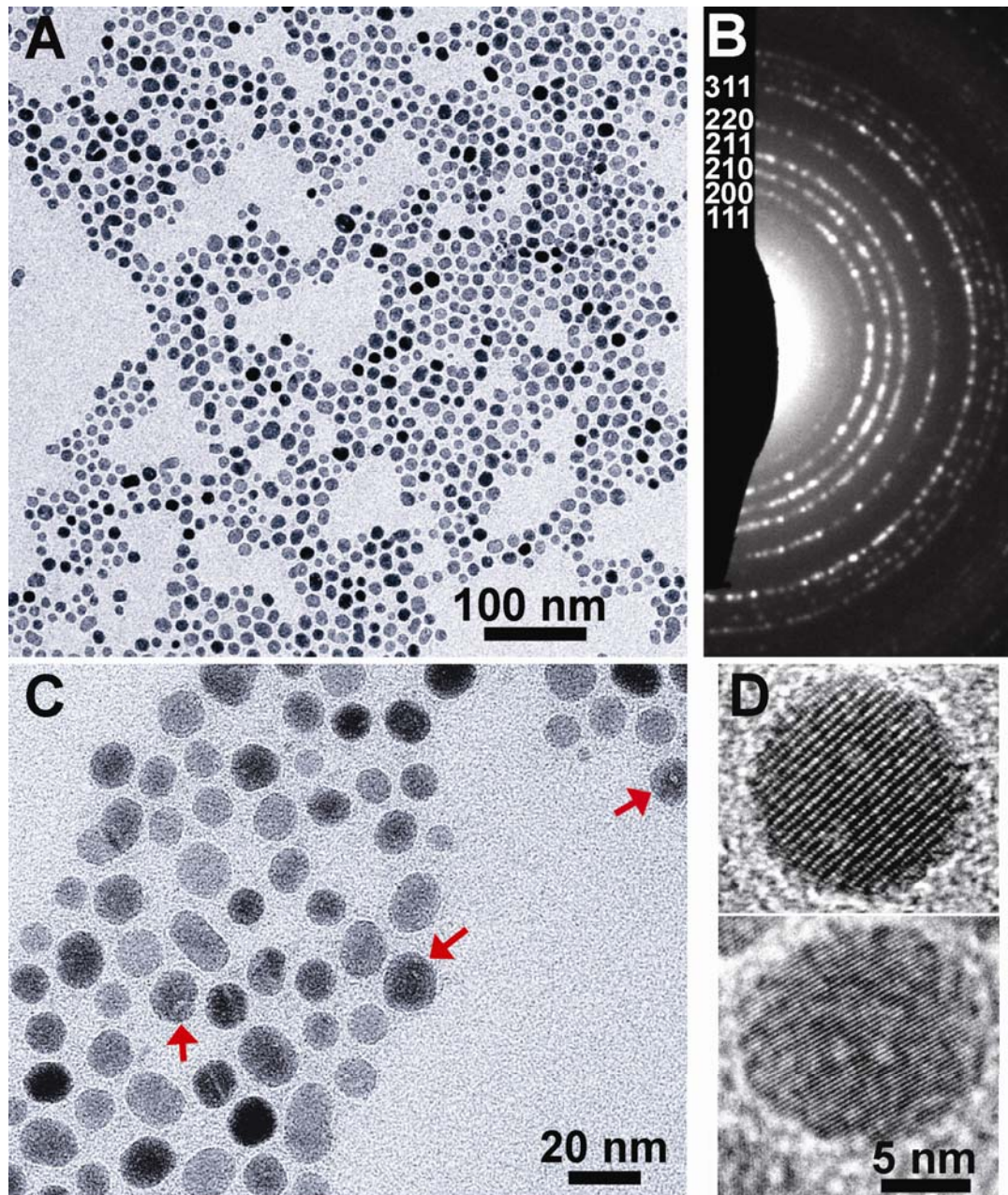
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<b>Pyrite</b> ( $\text{FeS}_2$ )	cubic, $E_g = 0.95$ eV (indirect) Van Vleck paramagnetic semiconductor
<b>Marcasite</b> ( $\text{FeS}_2$ )	orthorhombic, $E_g = 0.4$ eV (indirect) diamagnetic semiconductor
<b>Greigite</b> ( $\text{Fe}_3\text{S}_4$ )	cubic, $E_g = 0.4$ eV, ferrimagnetic
<b>Pyrrhotite</b> ( $\text{Fe}_{1-x}\text{S}$ , with $x \leq 0.2$ )	hexagonal or monoclinic, $E_g \approx 0.2$ eV diamagnetic or ferromagnetic semiconductor
<b>Troilite</b> ( $\text{FeS}$ )	hexagonal, $E_g = 0.04$ eV, antiferromagnetic semiconductor

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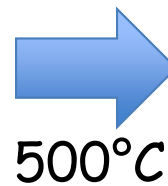
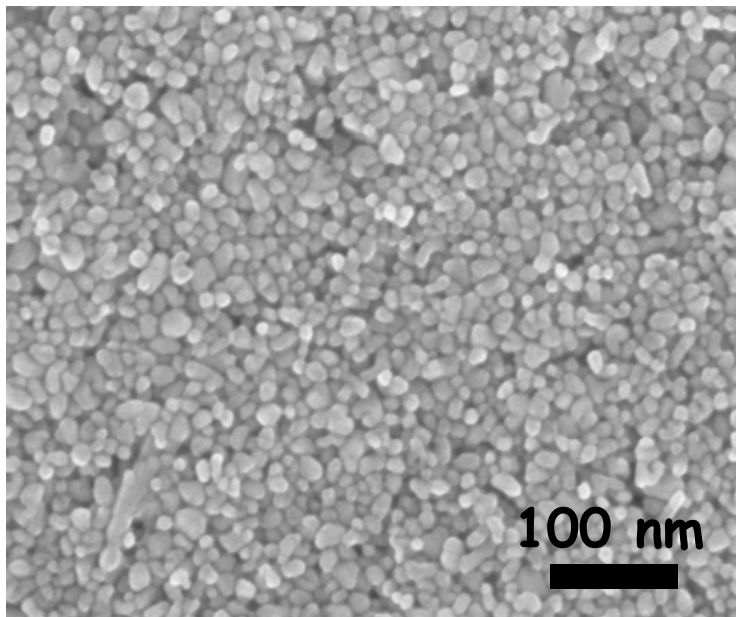


# Pyrite nanocrystal inks

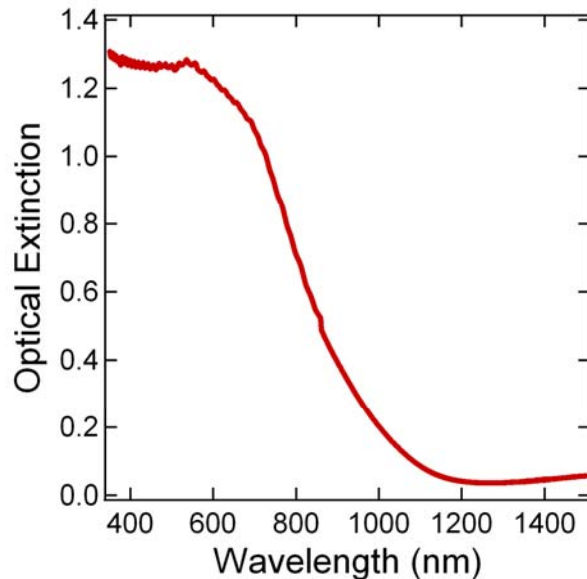
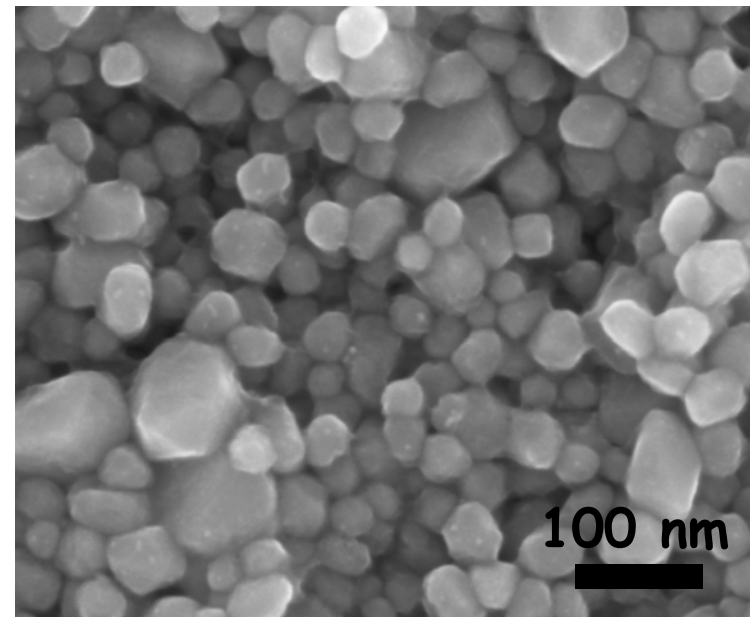


# Polycrystalline pyrite thin films

Sintering the NC films in  $S_2$  gas gives polycrystalline pyrite



500°C



- film optimization
- characterization
- surface passivation
- devices